

2. RECIPROCAL SPACE IN CRYSTAL-STRUCTURE DETERMINATION

the origin along the diad axis by restricting τ_1 to the family of vectors $\{\tau_1\}$ of type $[x0z]$. The practical consequence is that $\langle |F_h|^2 \rangle$ is significantly modified in polar space groups if h satisfies

$$h \cdot \tau_1 = 0,$$

where τ_1 belongs to the family of restricted vectors $\{\tau_1\}$.

(d) Atomic groups correctly positioned. Then (Main, 1976; Giacovazzo, 1983a)

$$\langle |F_h|^2 \rangle = |F_{p,h}|^2 + \varepsilon_h \sum_q,$$

where $F_{p,h}$ is the structure factor of the partial known structure and q are the atoms with unknown positions.

(e) A pseudotranslational symmetry is present. Let u_1, u_2, u_3, \dots be the pseudotranslation vectors of order n_1, n_2, n_3, \dots , respectively. Furthermore, let p be the number of atoms (symmetry equivalents included) whose positions are related by pseudotranslational symmetry and q the number of atoms (symmetry equivalents included) whose positions are not related by any pseudotranslation. Then (Casarano *et al.*, 1985a,b)

$$\langle |F_h|^2 \rangle = \varepsilon_h (\zeta_h \sum_p + \sum_q),$$

where

$$\zeta_h = \frac{(n_1 n_2 n_3 \dots) \gamma_h}{m}$$

and γ_h is the number of times for which algebraic congruences

$$h \cdot R_s u_i \equiv 0 \pmod{1} \quad \text{for } i = 1, 2, 3, \dots$$

are simultaneously satisfied when s varies from 1 to m . If $\gamma_h = 0$ then F_h is said to be a *superstructure reflection*, otherwise it is a *substructure reflection*.

Often substructures are not ideal: *e.g.* atoms related by pseudotranslational symmetry are ideally located but of different type (replacive deviations from ideality); or they are equal but not ideally located (displacive deviations); or a combination of the two situations occurs. In these cases a correlation exists between the substructure and the superstructure. It has been shown (Mackay, 1953; Casarano *et al.*, 1988a) that the scattering power of the substructural part may be estimated *via* a statistical analysis of diffraction data for ideal pseudotranslational symmetry or for displacive deviations from it, while it is not estimable in the case of replacive deviations.

2.2.4.2. Definition of quasi-normalized structure factor

When probability theory is not used, the *quasi-normalized structure factors* ε_h and the *unitary structure factors* U_h are often used. ε_h and U_h are defined according to

$$|\varepsilon_h|^2 = \varepsilon_h |E_h|^2$$

$$U_h = F_h / \left(\sum_{j=1}^N f_j \right).$$

Since $\sum_{j=1}^N f_j$ is the largest possible value for F_h , U_h represents the fraction of F_h with respect to its largest possible value. Therefore

$$0 \leq |U_h| \leq 1.$$

If atoms are equal, then $U_h = \varepsilon_h / N^{1/2}$.

2.2.4.3. The calculation of normalized structure factors

N.s.f.'s cannot be calculated by applying (2.2.4.1) to observed s.f.'s because: (a) the observed magnitudes I_h (already corrected for Lp factor, absorption, . . .) are on a relative scale; (b) $\langle |F_h|^2 \rangle$ cannot

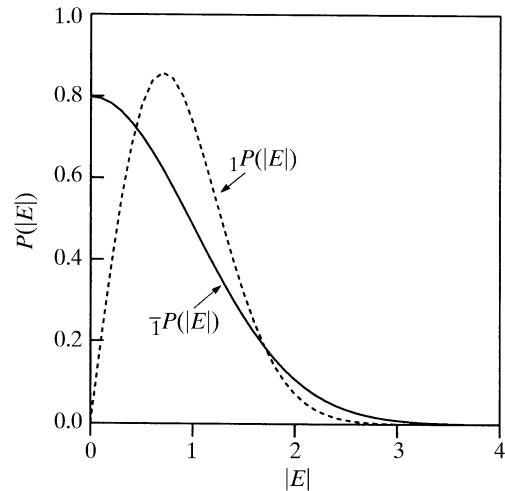


Fig. 2.2.4.1. Probability density functions for cs. and ncs. crystals.

be calculated without having estimated the vibrational motion of the atoms.

This is usually obtained by the well known Wilson plot (Wilson, 1942), according to which observed data are divided into ranges of $s^2 = \sin^2 \theta / \lambda^2$ and averages of intensity $\langle I_h \rangle$ are taken in each shell. Reflection multiplicities and other effects of space-group symmetry on intensities must be taken into account when such averages are calculated. The shells are symmetrically overlapped in order to reduce statistical fluctuations and are restricted so that the number of reflections in each shell is reasonably large. For each shell

$$K \langle I \rangle = \langle |F|^2 \rangle = \langle |F^o|^2 \rangle \exp(-2Bs^2) \quad (2.2.4.3)$$

should be obtained, where K is the scale factor needed to place X-ray intensities on the absolute scale, B is the overall thermal parameter and $\langle |F^o|^2 \rangle$ is the expected value of $|F|^2$ in which it is assumed that all the atoms are at rest. $\langle |F^o|^2 \rangle$ depends upon the structural information that is available (see Section 2.2.4.1 for some examples).

Equation (2.2.4.3) may be rewritten as

$$\ln \left\{ \frac{\langle I \rangle}{\langle |F^o|^2 \rangle} \right\} = -\ln K - 2Bs^2,$$

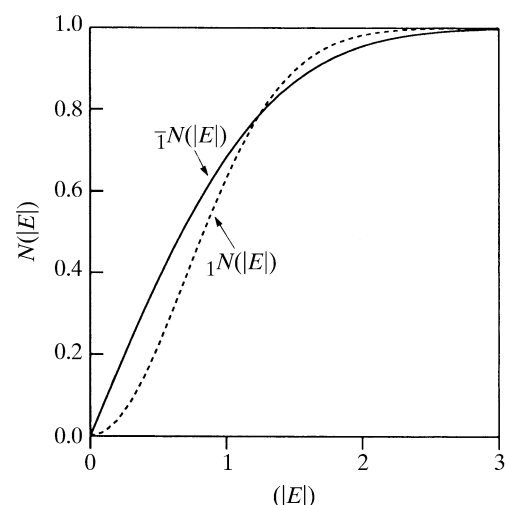


Fig. 2.2.4.2. Cumulative distribution functions for cs. and ncs. crystals.