

3.6. WHOLE POWDER PATTERN MODELLING

For cuprite, the literature (Tromans & Meech, 2001) suggests that the main slip system is (001){100}. The contrast factor can be calculated analytically from the single-crystal elastic constants of cuprite ($c_{11} = 121$, $c_{12} = 105$ and $c_{44} = 12.1$ GPa; Every & McCurdy, 1992b) following Martinez-Garcia *et al.* (2007):

$$\bar{C}_{\text{Cu}_2\text{O},e} = 0.355963 - 0.609491 \frac{h^2k^2 + k^2l^2 + h^2l^2}{(h^2 + k^2 + l^2)^2}, \quad (3.6.58)$$

$$\bar{C}_{\text{Cu}_2\text{O},s} = \frac{2h^2k^2 + k^2l^2 + h^2l^2}{3(h^2 + k^2 + l^2)^2}. \quad (3.6.59)$$

For tenorite, a different approach was followed. The phase is minor and the single-crystal elastic constants are not readily available: we can therefore use the contrast factor in an effective way by refining the coefficients of the corresponding invariant [see equation (3.6.42)]. This preserves the profile shape determined by Wilkens' theory and just dilutes the meaning of the dislocation density. The average contrast factor is

$$\begin{aligned} \bar{C}_{\text{CuO},(hkl)} &= \left\{ 4[E_1h^4 + E_2k^4 + E_3l^4 + 2(E_4h^2k^2 + E_5k^2l^2 + E_6h^2l^2) \right. \\ &\quad \left. + 4(E_7h^3k + E_8h^3l + E_9k^3h)]Y^4Z^4\sin^4\beta \right\} \\ &\quad \times \left(\{k^2Z^2 + 2Y^2(l^2 + h^2Z^2) - Z[4hlY^2\cos(\beta) + k^2Z\cos(2\beta)]\}^2 \right)^{-1}, \end{aligned} \quad (3.6.60)$$

where a , b , c and β are the unit-cell parameters of tenorite, $Y = b/a$ and $Z = c/a$.

The dislocation density in Cu_2O is quite high [$\rho = 2.8(5) \times 10^{16} \text{ m}^{-2}$]: dislocations are more of the edge character [$f_E = 0.85(3)$] and the outer cutoff radius $R_e = 9(3)$ nm leads to a Wilkens' parameter of approximately 1.5, suggesting a strong dislocation interaction. The high dislocation density in this material is justified by the very low shear modulus ($G = 10.3$ GPa; Every & McCurdy, 1992b), whereas the high dislocation interaction is the result of the severe deformation induced by the milling.

APPENDIX A3.6.1 Functions for profile shapes

The unit-area Gaussian $G(x, \omega)$ and Lorentzian $L(x, \omega)$ functions are defined as

$$G(x, \omega) = \frac{2\sqrt{\ln 2/\pi}}{\omega} \exp\left(-\frac{4x^2 \ln 2}{\omega^2}\right), \quad (3.6.61)$$

$$L(x, \omega) = \frac{2}{\pi\omega} \left(\frac{1}{1 + 4x^2/\omega^2} \right), \quad (3.6.62)$$

where x is the running variable and ω is the full-width at half-maximum. Based on these definitions, the Voigt and pseudo-Voigt are

$$V(x, \omega_L, \omega_G) = L(x, \omega_L) \otimes G(x, \omega_G) \quad (3.6.63)$$

and

$$pV(x, \omega_L, \omega_G) = \eta L(x, \omega_L) + (1 - \eta)G(x, \omega_G), \quad (3.6.64)$$

respectively, where η is the mixing parameter (ranging between 0 and 1) and ω_L and ω_G are the width of the Lorentzian and Gaussian components, respectively.

References

- Adler, T. & Houska, C. R. (1979). *Simplifications in the X-ray line-shape analysis*. *J. Appl. Phys.* **50**, 3282–3287.
- Alexander, L. (1954). *The synthesis of X-ray spectrometer line profiles with application to crystallite size measurements*. *J. Appl. Phys.* **25**, 155–161.
- Armstrong, N., Leoni, M. & Scardi, P. (2006). *Considerations concerning Wilkens' theory of dislocation line-broadening*. *Z. Kristallogr. Suppl.* **23**, 81–86.
- Balogh, L., Ribárik, G. & Ungár, T. (2006). *Stacking faults and twin boundaries in fcc crystals determined by X-ray diffraction profile analysis*. *J. Appl. Phys.* **100**, 023512.
- Balzar, D. & Popović, S. (1996). *Reliability of the simplified integral-breadth methods in diffraction line-broadening analysis*. *J. Appl. Cryst.* **29**, 16–23.
- Bergmann, J. & Kleeberg, R. (2001). *Fundamental parameters versus learnt profiles using the Rietveld program BGMN*. *Mater. Sci. Forum*, **378–381**, 30–37.
- Berkum, J. G. M. van (1994). *Strain Fields in Crystalline Materials*. PhD thesis, Technische Universiteit Delft, Delft, The Netherlands.
- Bertaut, E. F. (1949a). *Etude aux rayons X de la répartition des dimensions des cristallites dans une poudre cristalline*. *C. R. Acad. Sci.* **228**, 492–494.
- Bertaut, E. F. (1949b). *Signification de la dimension cristalline mesurée d'après la largeur de raie Debye-Scherrer*. *C. R. Acad. Sci.* **228**, 187–189.
- Bertaut, E. F. (1950). *Raies de Debye-Scherrer et répartition des dimensions des domaines de Bragg dans les poudres polycristallines*. *Acta Cryst.* **3**, 14–18.
- Beyerlein, K. R., Leoni, M. & Scardi, P. (2012). *Temperature diffuse scattering of nanocrystals*. *Acta Cryst.* **A68**, 382–392.
- Billinge, S. J. L. (2008). *Local structure from total scattering and atomic pair distribution function (PDF) analysis*. In *Powder Diffraction: Theory and Practice*, edited by R. E. Dinnebier & S. J. L. Billinge. London: Royal Society of Chemistry.
- Brese, N. E., O'Keeffe, M., Ramakrishna, B. L. & Von Dreele, R. B. (1990). *Low-temperature structures of CuO and AgO and their relationships to those of MgO and PdO*. *J. Solid State Chem.* **89**, 184–190.
- Bruker (2009). *DIFFRAC.SUITE TOPAS, Total Pattern Analysis Solution*. Version 5. Bruker AXS, Karlsruhe, Germany.
- Caglioti, G., Paoletti, A. & Ricci, F. P. (1958). *Choice of collimator for a crystal spectrometer for neutron diffraction*. *Nucl. Instrum.* **3**, 223–228.
- Cervellino, A., Giannini, C. & Guagliardi, A. (2003). *Determination of nanoparticle structure type, size and strain distribution from X-ray data for monatomic f.c.c.-derived non-crystallographic nanoclusters*. *J. Appl. Cryst.* **36**, 1148–1158.
- Cheary, R. W. & Coelho, A. (1992). *A fundamental parameters approach to X-ray line-profile fitting*. *J. Appl. Cryst.* **25**, 109–121.
- Cheary, R. W. & Coelho, A. (1994). *Synthesizing and fitting linear position-sensitive detector step-scanned line profiles*. *J. Appl. Cryst.* **27**, 673–681.
- Cheary, R. W. & Coelho, A. A. (1998a). *Axial divergence in a conventional X-ray powder diffractometer. I. Theoretical foundations*. *J. Appl. Cryst.* **31**, 851–861.
- Cheary, R. W. & Coelho, A. A. (1998b). *Axial divergence in a conventional X-ray powder diffractometer. II. Realization and evaluation in a fundamental-parameter profile fitting procedure*. *J. Appl. Cryst.* **31**, 862–868.
- Cline, J. P., Black, D., Windover, D. & Henins, A. (2010). *SRM 660b – Line Position and Line Shape Standard for Powder Diffraction*. https://www.nist.gov/srmors/view_detail.cfm?srm=660b.
- Coelho, A. A. (2005). *A bound constrained conjugate gradient solution method as applied to crystallographic refinement problems*. *J. Appl. Cryst.* **38**, 455–461.
- Coelho, A. A. (2009). *TOPAS Academic*. Version 5. <http://www.topas-academic.net/>.
- Cozzoli, P. D., Snoeck, E., Garcia, M. A., Giannini, C., Guagliardi, A., Cervellino, A., Gozzo, F., Hernando, A., Achterhold, K., Ciobanu, N., Parak, F. G., Cingolani, R. & Manna, L. (2006). *Colloidal synthesis and characterization of tetrapod-shaped magnetic nanocrystals*. *Nano Lett.* **6**, 1966–1972.
- Debye, P. (1915). *Zerstreuung von Röntgenstrahlen*. *Ann. Phys.* **351**, 809–823.